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Drying Shrinkage, Strength, and Microstructure of Alkali-Activated High-Calcium Fly Ash using FGD-gypsum and Dolomite as Expansive Additive

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Abstract

This paper aims to investigate drying shrinkage, strength development, and microstructure of alkali-activated high-calcium fly ash (AAFA) paste by using FGD-gypsum

(FGD) and dolomite (DLM) as the expansive additive. The 10 molar of sodium hydroxide combined with sodium silicate solutions were used as liquid alkaline activation in all mixtures. The preparation of the AAFA paste was divided into 2 steps. First, the study was focused on the effects of expansive additive (FGD and DLM) and liquid/binder (L/B) ratio on the drying shrinkage of the AAFA paste. The FA was replaced by FGD and DLM at the dosages of 0%, 2.5%, 5.0%, 7.5%, and 10% by weight of binder. The AAFA paste with low drying shrinkage at the age of 90 days would be selected to further study. The other was to study setting time, compressive strength, and microstructure of the AAFA paste containing FGD and DLM. Test results indicated that the use of FGD and DLM as the expansive additive could improve the drying shrinkage of the AAFA paste. The increasing L/B ratio had adverse effects on its drying shrinkage. The setting time of the AAFA paste containing FGD and DLM had decreased, whereas its strength development had increased. However, the strength development of the AAFA paste incorporated with FGD tended to decline at the curing time of 120 days. These behaviors were consistent with the results of XRD, FTIR, and SEM/EDS analyses. The differences in reaction products were an essential factor in the strength development of the AAFA paste with expansive additive. It can be recommended that using DLM was advantages regarding the improvement of drying shrinkage and long-term strength in comparison with the AAFA paste with FGD.

Keywords: Alkali-activated binder, High-calcium fly ash, Expansive additive, Drying shrinkage, Strength development, Microstructure

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1. Introduction

Nowadays, Portland cement (PC) is a popular choice for use as a cementitious material in the construction industry. This is because the PC provided good mechanical properties and low-cost advantages. Besides, this PC could be used as the precursor for manufacturing the polymer-modified cement for repairing damaged concrete structures [1]. However, it is well known that PC is a primary problem of global warming, which causes a high amount of CO₂ emission. Turner and Collins [2] and Mclellan et al. [3] reported that every 1 ton of PC released approximately 1 ton of CO₂. This is why the amount of greenhouse gas in the atmosphere tends to increase every year. To solve this issue, a number of researchers [4-6] attempted to reduce the PC content for producing the concrete. Sata et al. [4] studied on the use of by-product viz., ground pulverized coal combustion fly ash, ground fluidized bed combustion fly ash, ground rice husk ash, and ground palm oil fuel ash to investigate mechanical properties of high-strength concrete. They reported that these pozzolanic materials could enhance the strength development of concrete. Detphan et al. [6] reported that the use of high-calcium fly ash (FA) and expanded perlite to replace PC could improve the strength and water retention requirements for masonry and rendering mortar as per ASTM standard. However, the PC is still widely used for producing the mortar and concrete. Therefore, the new eco-friendly cement is extensively studied in order to reduce the CO₂ footprint. One of them is the alkali-activated binders that could produce from precursors consisting of Si⁴⁺ and Al³⁺ ions activated with a high alkali solution [7, 8]. Many researchers [9-12] reported that the alkali-activated binders provided excellent mechanical and durability properties. Besides, alkali-activated binders showed a lower CO₂ footprint than those of PC and epoxy-resin.

Pacheco-Torgal et al. [13] revealed that the alkali-activated binders could be classified into two systems: low- and high- calcium system. The former, also known as geopolymers, is generally obtained from the activation of low-calcium materials and high alkaline solutions.

The sodium aluminosilicate hydrate (N-A-S-H) gel is the main reaction product, in which low strength when cured at ambient temperature was obtained [14]. For the latter, precursors such as slag and high-calcium FA activated with medium alkali solutions. The main reaction products of high-calcium based are calcium silicate hydrate (C-S-H) and calcium aluminosilicate hydrate (C-A-S-H) coexisted with sodium aluminosilicate hydrate (N-A-S-H) gel [15]. The occurrence of C-S-H in the alkali-activated binders is suitable to use in real construction because it could be provided the high strength when cured at ambient temperature.

In Thailand, high-calcium FA from Mae Moh power station is widely used for producing alkali-activated binders. The studies of Phoo-ngernkham et al. [16] and Phoo-ngernkham et al. [17] have shown that the use of FA with additives exhibited the early strength development like the PC system. Thus, it could be used as repair material described in previous studies [10-12, 18]. Note that the properties of repair material in terms of rapid setting, high strength, high bonding, and high durability are needed. However, some researchers [19, 20] reported that the shrinkage behavior of alkali-activated high-calcium FA showed a very high value when cured at ambient temperature. Ma et al. [21] have found that the shrinkage behavior is a critical property that caused the cracking probability of alkali-activated binders under the restrained condition, and hence long term durability problems occurred. Therefore, this may be a limitation of using alkali-activated high-calcium FA as the repair material, although it exhibited high bonding and excellent durability. Generally, alkali-activated binders had much higher autogenous and drying shrinkage than the PC system [21]. However, the test results of Riddirud et al. [19] and Fernandez-Jimenez et al. [22] showed that the drying shrinkage of heat-cured alkali-activated binders was much lower than that of PC, indicating that different curing conditions could influence the drying shrinkage of alkali-activated binders. The improvement of drying shrinkage of alkali-activated binders has been

studied by several researchers. Matalkah et al. [23] improved the drying shrinkage of alkali-activated cement by using 5%wt silica fume, 2.5%wt Polyethylene glycol, 5%wt gypsum, and 5%wt limestone powder. Bakharev et al. [24] reported that using 6% by weight of gypsum could reduce both autogenous and drying shrinkage of alkali-activated slag because the formation of ettringite acted as the expansive phase. Besides, the use of carbonate mineral viz., calcite and dolomite could reduce the drying shrinkage of low-calcium based geopolymers as reported by Yip et al. [25] and Cohen et al. [26]. Their results revealed that the formations of calcium carbonate (CaCO_3), magnesium hydroxide (Mg(OH)_2) and sodium carbonate (Na_2CO_3) were taken place and hence its drying shrinkage reduced. Hwang et al. [27] and Jin et al. [28] also investigated the shrinkage compensation of alkali-activated slag using reactive MgO as the expansive additive. It is found that using MgO in alkali-activated slag produced the reaction products that acted as the expansive phase. In addition, Punurai et al. [20] improved the drying shrinkage of alkali-activated high-calcium FA by using basalt fiber.

From the above review, there is no research investigated on the use of FGD-gypsum and dolomite as the expansive additive to improve the drying shrinkage of alkali-activated high-calcium FA. Thus, this research focuses mainly on the utilization of FGD-gypsum and dolomite as the expansive additive for improving the drying shrinkage and strength development of alkali-activated high-calcium FA. The obtained knowledge of this study would be very beneficial to understand the shrinkage compensation of alkali-activated high-calcium FA by using FGD-gypsum and dolomite as the expansive additive.

2. Experimental details

2.1 Materials

High-calcium FA from Mae Moh power plant in northern Thailand was used as the main precursor to produce alkali-activated binder. FA had specific gravity and median particle size of 2.65 and 15.3 μm , respectively. The FGD-gypsum (FGD) and dolomite (DLM) were used as the expansive additive (EA) for shrinkage compensation material. FGD is the by-product from Mae Moh power plant in northern Thailand, whereas the DLM is an anhydrous carbonate mineral. Both of FGD and DLM were oven-dried at 100°C for 24 hours and then passed through a sieve No. 100 (150 μm) before used as the precursor. FGD and DLM had specific gravity of 2.64 and 2.79, respectively. The chemical compositions of FA, FGD and DLM are shown in Table 1. The FA had sum of $\text{SiO}_2 + \text{Al}_2\text{O}_3 + \text{Fe}_2\text{O}_3$ at 66.94% and CaO at 21.41%. Therefore, this FA was classified as class C FA as per ASTM C618 [29]. The major components of FGD are CaO and SO_3 at 32.84% and 48.47%, whereas the DLM mainly consists of SiO_2 and CaO at 10.64% and 44.55%, respectively. Figure 1 shows the mineral compositions of the FGD and DLM as determined by XRD analysis. The FGD consists of crystal of gypsum ($\text{CaSO}_4 \cdot 2\text{H}_2\text{O}$) and calcium sulfate hemihydrate ($\text{CaSO}_4 \cdot 0.5\text{H}_2\text{O}$), whereas the DLM contains crystal phase of calcium carbonate (CaCO_3) and dolomite ($\text{CaMg}(\text{CO}_3)_2$). In addition, the SEM images of FGD and DLM are shown in Figure 2.

Table 1 Chemical compositions of FA, FGD, and DLM (by weight)

Materials	Chemical compositions (%)											
	SiO_2	Al_2O_3	Fe_2O_3	CaO	MgO	K_2O	Na_2O	TiO_2	P_2O_5	other	SO_3	LOI
FA	36.93	18.10	11.91	21.41	2.78	2.28	1.42	0.36	0.20	0.16	2.90	1.54
FGD	0.73	0.42	0.09	32.84	0.76	0.01	0.17	0.03	0.04	0.01	48.47	16.44
DLM	10.64	1.19	0.47	44.55	3.09	0.15	0.03	0.06	0.05	0.06	0.13	39.57

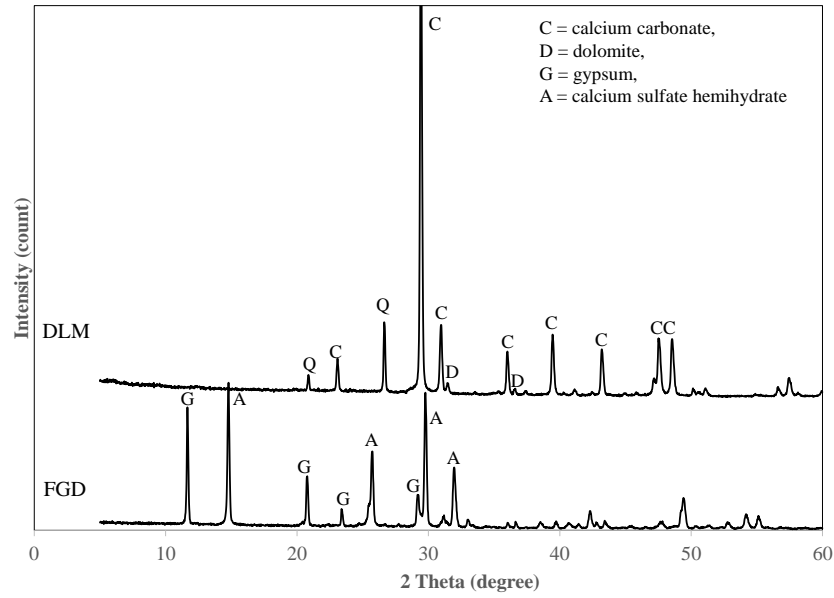
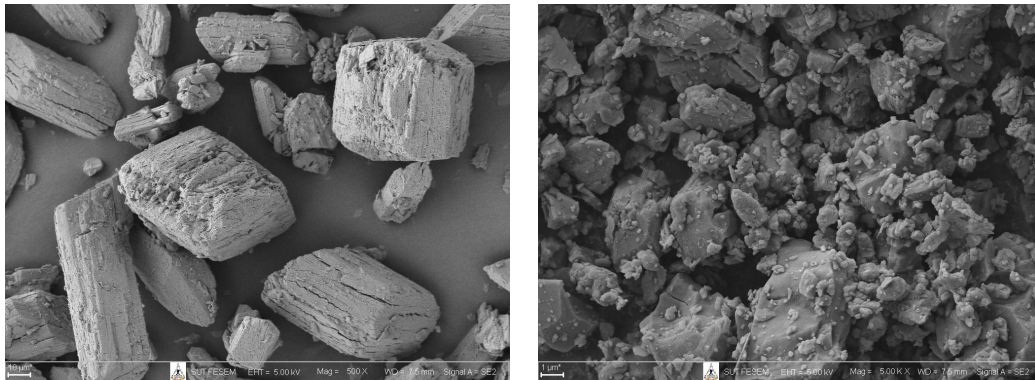


Figure 1 XRD patterns of FGD and DLM



(a) FGD

(b) DLM

Figure 2 SEM images of FGD and DLM

2.2 Mix proportions

The mix proportions of alkali-activated FA (AAFA) paste with expansive additive are summarized in Table 2. The 10 molar of sodium hydroxide (NaOH) and sodium silicate (Na_2SiO_3) with 13.89% Na_2O , 32.15% SiO_2 , and 53.96% H_2O were used as liquid activators with constant $\text{Na}_2\text{SiO}_3/\text{NaOH}$ ratio of 1.0. In this study, preparation of the AAFA paste with the expansive additive was divided into 2 steps as follows:

(1) The expansive additive (FGD and DLM) was used to replace FA at the amount of 0%, 2.5%, 5.0%, 7.5%, and 10% by weight of binder. The differences in liquid/binder (L/B) ratios of 0.40, 0.50, and 0.60 were investigated. After testing, the best result of 90-day drying shrinkage of the AAFA paste with expansive additive (FGD and DLM) was considered for conducting the next step.

(2) As mentioned, the L/B ratio of the AAFA paste with the expansive additive is chosen regarding the lowest drying shrinkage. Then, the setting time, compressive strength, and microstructure of the AAFA paste with expansive additive are tested and examined.

Table 2 Mix proportions of alkali-activated FA paste with expansive additive

No.	Symbol	FA (g)	EA (g)	NaOH (g)	Na ₂ SiO ₃ (g)
1	0.4LB100FA	100.0	-	20	20
2	0.4LB2.5EA	97.5	2.5	20	20
3	0.4LB5.0EA	95.0	5.0	20	20
4	0.4LB7.5EA	92.5	7.5	20	20
5	0.4LB10EA	90.0	10.0	20	20
6	0.5LB100FA	100.0	-	25	25
7	0.5LB2.5EA	97.5	2.5	25	25
8	0.5LB5.0EA	95.0	5.0	25	25
9	0.5LB7.5EA	92.5	7.5	25	25
10	0.5LB10EA	90.0	10.0	25	25
11	0.6LB100FA	100.0	-	30	30
12	0.6LB2.5EA	97.5	2.5	30	30
13	0.6LB5.0EA	95.0	5.0	30	30
14	0.6LB7.5EA	92.5	7.5	30	30
15	0.6LB10EA	90.0	10.0	30	30

For the mixing of the paste, NaOH and Na₂SiO₃ solutions were firstly mixed together before use as the liquid alkaline activators. The FA and expansive additive (FGD and DLM)

were dry mixed until the mixture was almost homogenous. After that, liquid alkaline activators was added into the mixture and the mixing was done for 2 minutes.

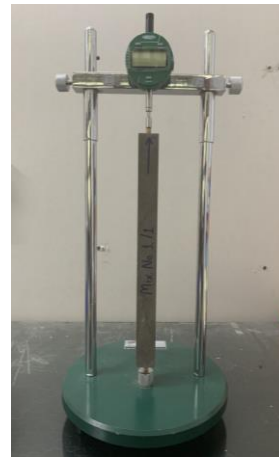
2.3 Testing procedure

2.3.1 Drying shrinkage

After mixing, a fresh AAFA paste was placed into a 25x25x285 mm³ prism mold following ASTM C490 [30] and ASTM C596 [31]. The shrinkage specimens were demolded after ambient temperature curing for 24 hours and then they were measured as the first length. Then, the specimens were kept in a control room of 25±2°C with a relative humidity of 50±5%. The length changes were measured daily for the first 2 weeks and then once a week until a month. After that, they were measured every month until 360 days. The test set-up of drying shrinkage specimens using digital micrometer apparatus is shown in Figure 3.



(a) specimens



(b) Length change test apparatus

Figure 3 Test set-up of drying shrinkage specimens

2.3.2 Setting time and compressive strength

The setting time of the AAFA paste with expansive additive was evaluated using the Vicat apparatus in accordance with ASTM C191 [32]. For the compressive strength test, the

fresh AAFA pastes were placed into a 50x50x50 mm³ cube mold and compacted as per ASTM C109 [33]. After that, a cube mold was immediately wrapped with vinyl sheet to protect moisture loss for 24 hours. They were demolded and then immediately wrapped using vinyl sheet and kept in ambient temperature until testing ages of 7, 28, 60, 90, 120, and 180 days. Five samples were tested for each case and their average value was used as the reported experimental result.

2.3.3 XRD, SEM, FTIR analyses

To study the evolution of the growth of cementitious products in AAFA paste with expansive additive, microstructure evaluation techniques viz., X-ray diffractometry (XRD), scanning electron microscopic (SEM), Fourier transform infrared spectroscopy (FTIR), were conducted. After the compressive strength was done at the age of 90 days, the specimens were broken into small pieces approximately 3-6 mm for SEM analysis. Besides, the specimens were ground into fine powder for XRD and FTIR analyses. The 2theta between 10 and 60° were performed to analyze the phase formation in the XRD [34], whereas the wave number region between 400 and 4000 cm⁻¹ was conducted for the FTIR.

3. Results and Discussion

3.1 Drying shrinkage

Figure 4 shows the drying shrinkage of the AAFA paste using FGD and DLM as the expansive additive under different L/B ratios up to 360 days. Generally, the drying shrinkage of alkali-activated pastes is significant for tensile stress developed in the matrix [21]; therefore, a high drying shrinkage was related to the possibility of cracking for the samples. According to Figures 4-9, the drying shrinkage of the AAFA paste is remarkably reduced as the replacement levels of FGD and DLM increased. For example, the 90-day drying

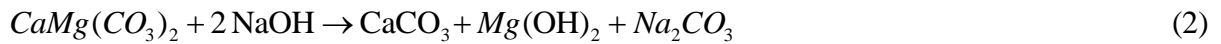
shrinkage of 0.40LB100FA, 0.40LB2.5FGD, 0.40LB5.0FGD, and 0.40LB7.5FGD pastes are 12,076, 10,460, 9,422, and $9,512 \times 10^{-6}$ mm/mm, whereas the 90-day drying shrinkage of 0.40LB2.5DLM, 0.40LB5.0DLM, 0.40LB7.5DLM and 0.40LB10DLM pastes are 9,756, 8,656, 8,372, and $12,484 \times 10^{-6}$ mm/mm, respectively. The high values of drying shrinkage of the AAFA paste are consistent with what was reported by Punurai et al. [20] and Yusuf et al. [35]. As reported by previous researchers, the 90-day drying shrinkage of FA geopolymer paste was approximately $30,000 \times 10^{-6}$ mm/mm [20], whereas the 90-day drying shrinkage of alkali-activated paste made from ground blast-furnace slag with ultra-fine palm oil fuel ash was approximately $26,000 \times 10^{-6}$ mm/mm [35].

According to Figures 4-9, the incorporation of FGD and DLM as the expansive additive is effective for compensating the drying shrinkage of the AAFA paste as expected. This is because of the expansion potential of FGD and DLM. As is shown in Table 1 and Figure 1, the FDG is mainly consisted of gypsum ($\text{CaSO}_4 \cdot 2\text{H}_2\text{O}$) and calcium sulfate hemihydrate ($\text{CaSO}_4 \cdot 0.5\text{H}_2\text{O}$), whereas the DLM mainly consisted of calcium carbonate (CaCO_3) and dolomite ($\text{CaMg}(\text{CO}_3)_2$). This agrees with the previous report [23] on the enhancement of drying shrinkage of alkali-activated binders cured at room temperature by using gypsum ($\text{CaSO}_4 \cdot 2\text{H}_2\text{O}$) and limestone (CaCO_3). Punurai et al. [20] and Pangdaeng et al. [36] also claimed that additional C-S-H and/or C-A-S-H co-existed with N-A-S-H gel within the matrix could lead to the refinement of pore structure [20].

Regarding the effect of the L/B ratio on the behavior of drying shrinkage, it is found that the increasing L/B ratios are found to increase the drying shrinkage of the AAFA paste for all replacement levels of FGD and DLM. For example, the 90-day drying shrinkage of 0.40LB100FA, 0.40LB5FGD, 0.50LB100FA, 0.50LB5FGD, 0.60LB100FA, and 0.60LB5FGD pastes are 12,076, 9,422, 12,409, 9,727, 15,827, and $13,976 \times 10^{-6}$ mm/mm, whereas the 90-day drying shrinkage of 0.40LB5DLM, 0.50LB5DLM, and 0.60LB5DLM

pastes are 8,656, 10,291, and 12,307 $\times 10^{-6}$ mm/mm, respectively. These results agreed with Ridtirud et al. [19] who reported that the excess of water content at a high L/B ratio, resulting in high porosity and high shrinkage. Moreover, a higher Na₂O and SiO₂ contents from NaOH and Na₂SiO₃ solutions exhibited a larger drying shrinkage of alkali-activated low-calcium FA [21].

A number of researchers had reported why using FGD and DLM could enhance the drying shrinkage of the AAFA paste. Chindaprasirt et al. [37] reported that the reaction of gypsum (CaSO₄.2H₂O) in alkali solution produced calcium hydroxide (Ca(OH)₂) and sodium sulfate (Na₂SO₄) as shown in Eq. (1). Chindaprasirt et al. [37] and Bakharev et al. [24] explained that ettringite formation (Ca₆Al₂(SO₄)₃(OH)₁₂.26H₂O) would compensate for the shrinkage behavior of alkali-activated binders. Ettringite formation rate depended on the concentration of OH⁻, SO₄²⁻, and Ca²⁺ in the liquid alkaline; therefore, less formation of ettringite was performed at high NaOH concentration [38]. According to Figures 4-6, the FA:FGD ratios at 95:5, 90:10, and 97.5:2.5 are found to be the lowest drying shrinkage at the L/B ratios of 0.40, 0.5, 0.60, respectively. Differences in the trend of the FGD replacement levels are due to the characteristic of FGD. Formations of ettringite and sodium sulfate (Na₂SO₄) are reasonable to gain low in drying shrinkage. As is observed in Figure 5, the use of 10% FGD activated with the L/B ratio of 0.50 provides low drying shrinkage.



Regarding the use of DLM in the AAFA paste, the dedolomitization reaction is illustrated in Eqs. (2) and (3) as reported by Garcia et al [39]. When dolomite ($\text{CaMg}(\text{CO}_3)_2$) activated with NaOH solution, formations of calcium carbonate (CaCO_3), magnesium hydroxide ($\text{Mg}(\text{OH})_2$) and sodium carbonate (Na_2CO_3) were formed. This is why the drying shrinkage of the AAFA paste could reduce except the paste with 10%DLM (see Figures 7-9), indicating that there is a turning point of the DLM addition. The high content of DLM corresponds to high magnesium hydroxide ($\text{Mg}(\text{OH})_2$) in the matrix [26]; hence, magnesium hydroxide ($\text{Mg}(\text{OH})_2$), known as “alkali-carbonate reaction”, caused the increasing volume and expansion of the hardened matrix. Besides, magnesium hydroxide ($\text{Mg}(\text{OH})_2$) reacts with CO_2 to form magnesium carbonate (MgCO_3) and H_2O as shown in Eq. (4) [40].

From the test results obtained, it can be summarized that the L/B ratio of 0.40 exhibits lowest drying shrinkage of the AAFA paste with FGD and DLM as the expansive additive at ranges between 90 to 360 days. Therefore, they were selected to investigate the experiments of setting time and compressive strength of the AAFA paste.

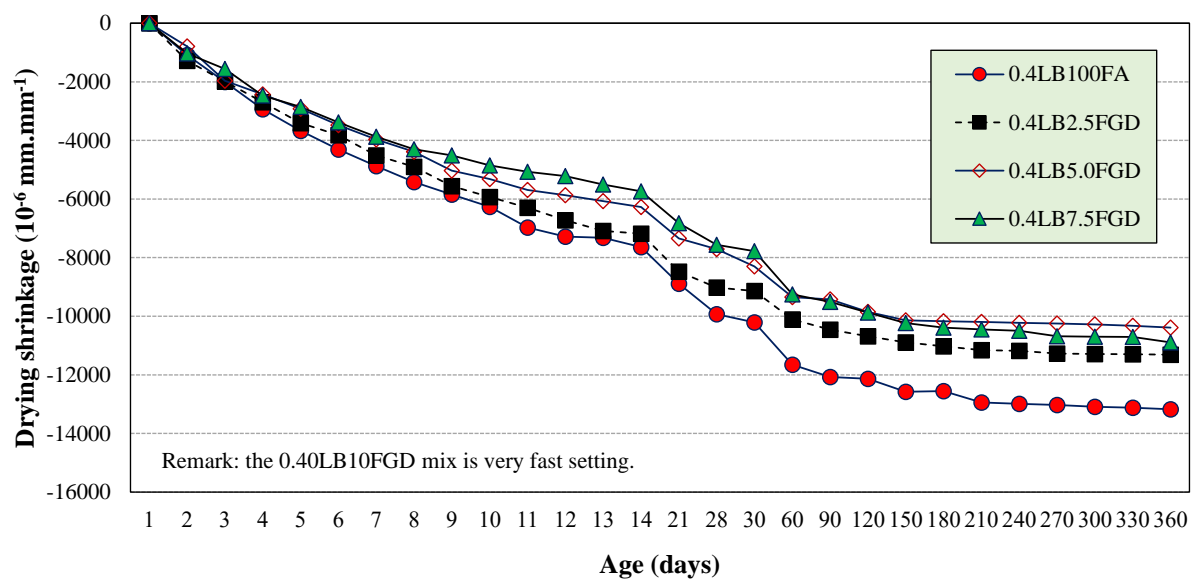


Figure 4 Drying shrinkage of alkali-activated FA paste at various FGD contents
for L/B of 0.40

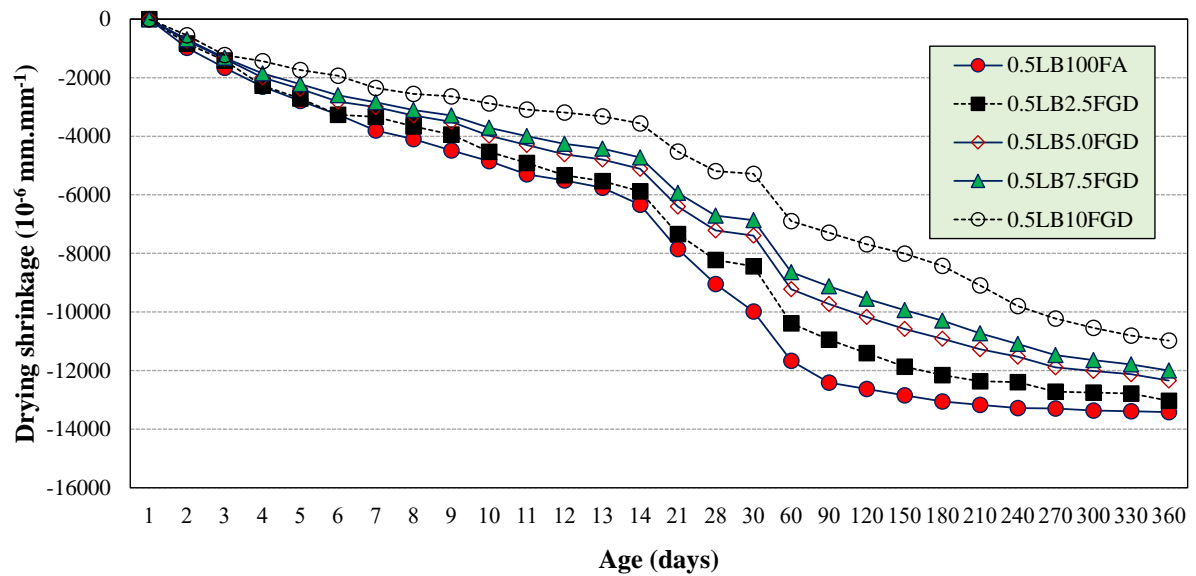


Figure 5 Drying shrinkage of alkali-activated FA paste at various FGD contents
for L/B of 0.50

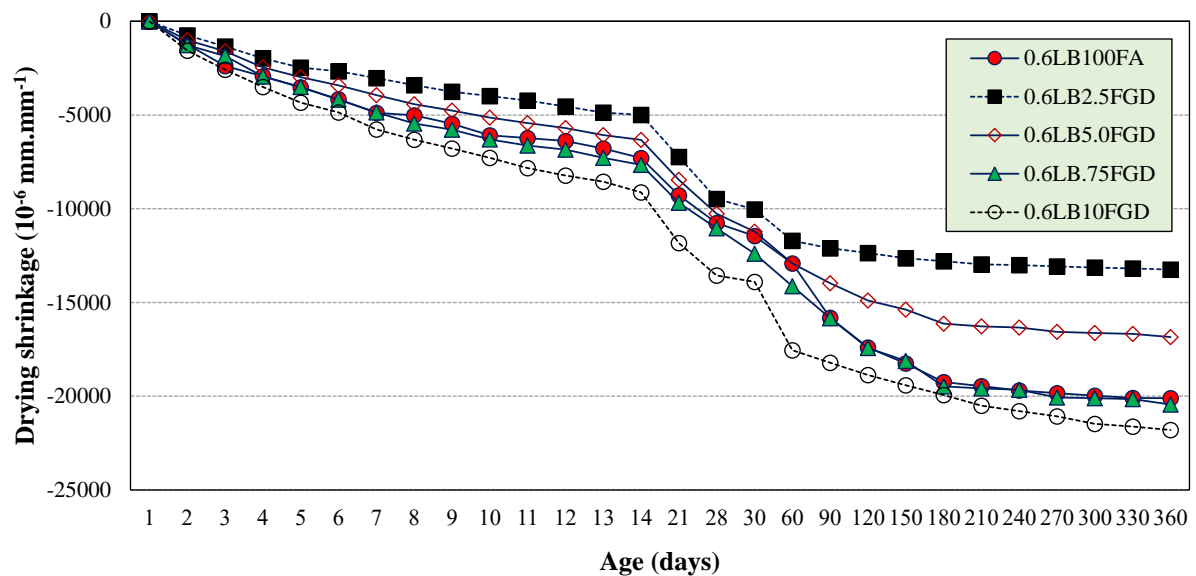


Figure 6 Drying shrinkage of alkali-activated FA paste at various FGD contents
for L/B of 0.60

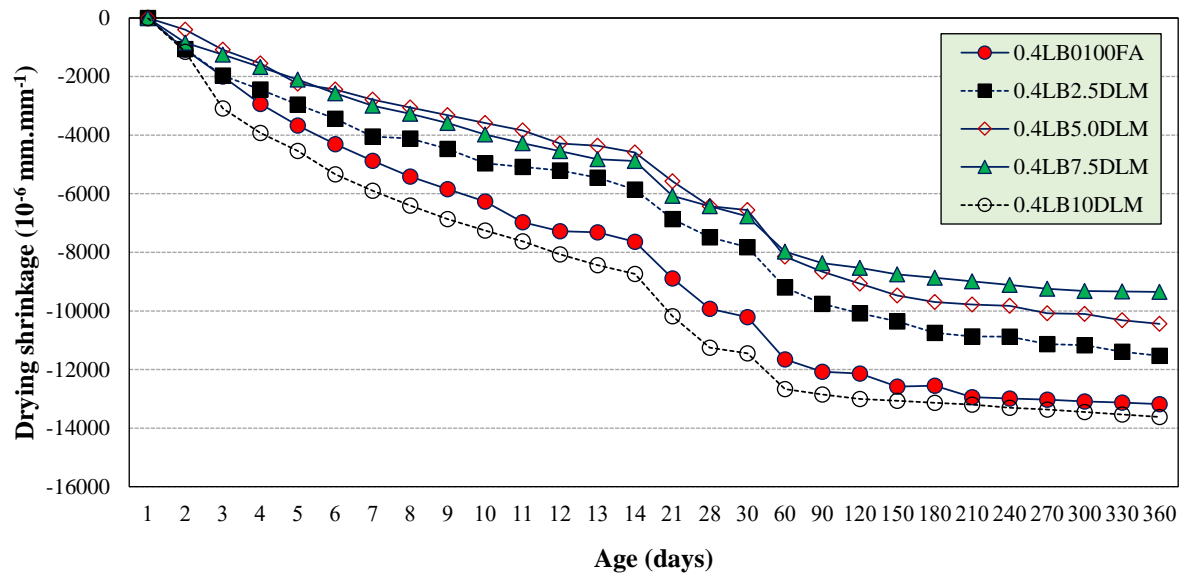


Figure 7 Drying shrinkage of alkali-activated FA paste at various DLM contents
for L/B of 0.40

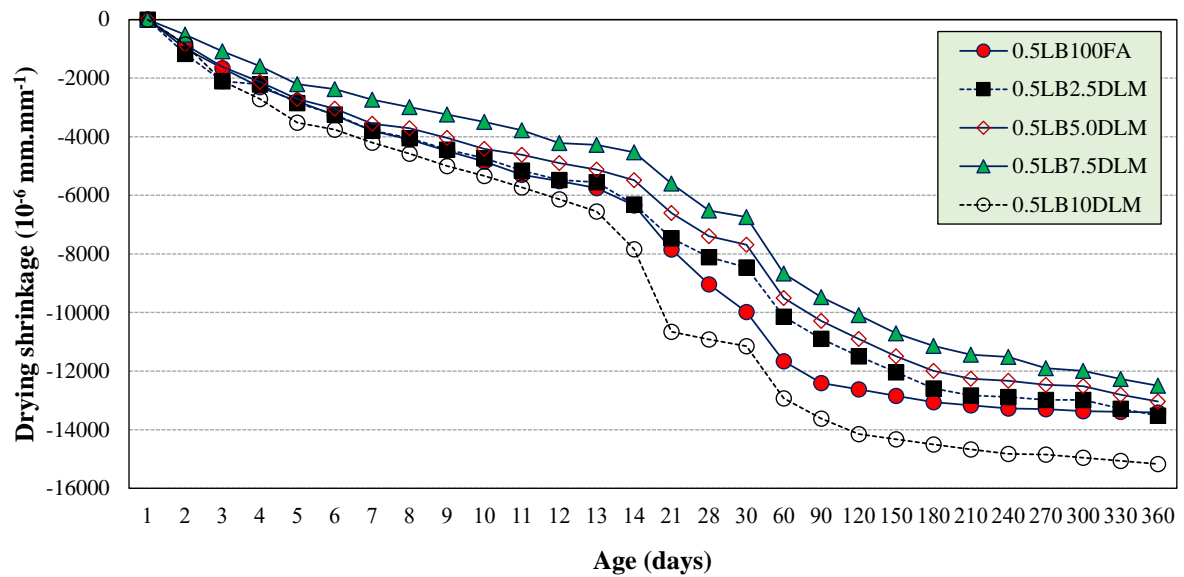


Figure 8 Drying shrinkage of alkali-activated FA paste at various DLM contents
for L/B of 0.50

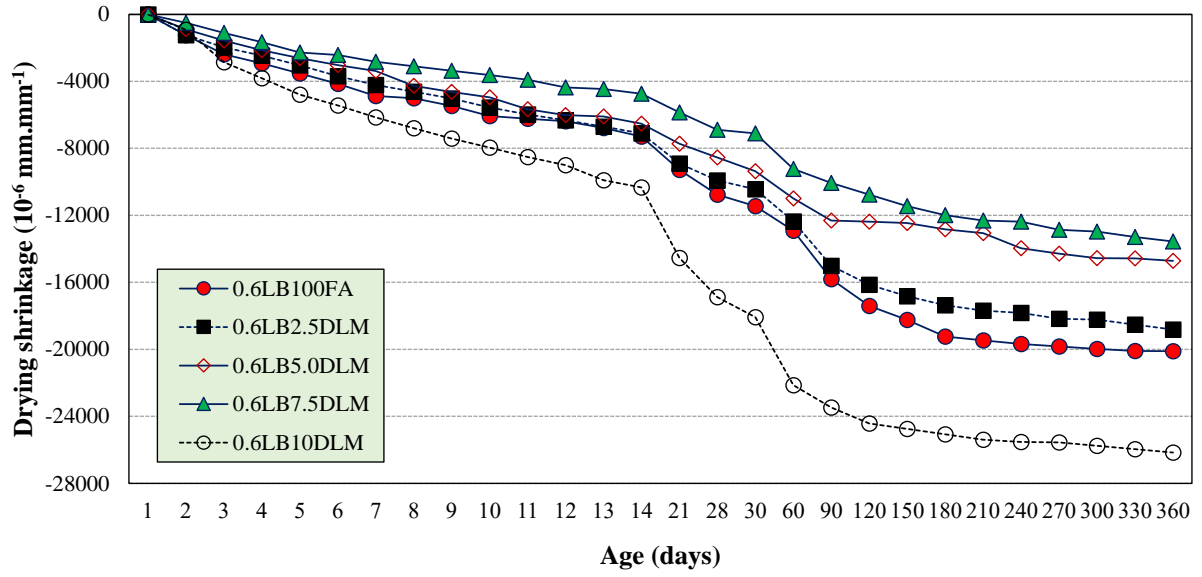


Figure 9 Drying shrinkage of alkali-activated FA paste at various DLM contents
for L/B of 0.60

3.2 Setting time

Figure 10 shows the change in the setting time of the AAFA paste as a function of FGD and DLM contents. The initial and final setting times of the AAFA paste obviously decrease with increasing content of FGD and DLM. They are 5-8 min and 9-13 min for the AAFA pastes with FGD, and 7-9 min and 10-14 min for the AAFA pastes with DLM. While the initial and final setting times of 100FA are 12 min and 18 min, respectively. The presence of Ca in the raw materials is an essential factor, which provides the formation of C-S-H within the matrix [34, 41]. A number of researchers, for example [12, 17, 36, 42-45], reported that the CaO was the main factor that generates heat within the matrix and hence accelerates the geopolymerization process. Chindaprasirt et al. [46] and Phoo-ngernkham et al. [43] demonstrated that the increase in CaO content corresponded to a decrease of the $\text{SiO}_2/\text{Al}_2\text{O}_3$ ratio. As mentioned in [47], this could lead to higher reaction degree within the matrix.

Comparing the use of FGD and DLM for producing the AAFA paste, using FGD exhibits the faster setting time of paste than that of DLM, implying that the $\text{CaSO}_4 \cdot 2\text{H}_2\text{O}$ from FGD had very high dissolution compared with that of $\text{CaMg}(\text{CO}_3)_2$ from DLM. Rattanasak et al. [48] claimed that the solubility constant (K_{sp}) of calcium hydroxide ($\text{Ca}(\text{OH})_2$) and calcium carbonate (CaCO_3) could explain the results of setting time and compressive strength. Note that the formations of calcium hydroxide ($\text{Ca}(\text{OH})_2$) and calcium carbonate (CaCO_3) were obtained by the reaction products as shown in Eqs. (1) and (2). As reported, K_{sp} of $\text{Ca}(\text{OH})_2$ was 7.9×10^{-6} , whereas K_{sp} of CaCO_3 was 3.8×10^{-6} ; therefore, higher K_{sp} value could accelerate the reaction degree within the paste. In addition, Rattanasak et al. [48] demonstrated that the reaction degree of $\text{Mg}(\text{OH})_2$ as obtained by Eq. (2) could dissolve lower than $\text{Ca}(\text{OH})_2$ as obtained by Eq. (1). Note that the fast setting of the AAFA pastes with FGD and DLM as the expansive additive is advantageous for repair material as specified by ASTM C881/C881M-14 [49].

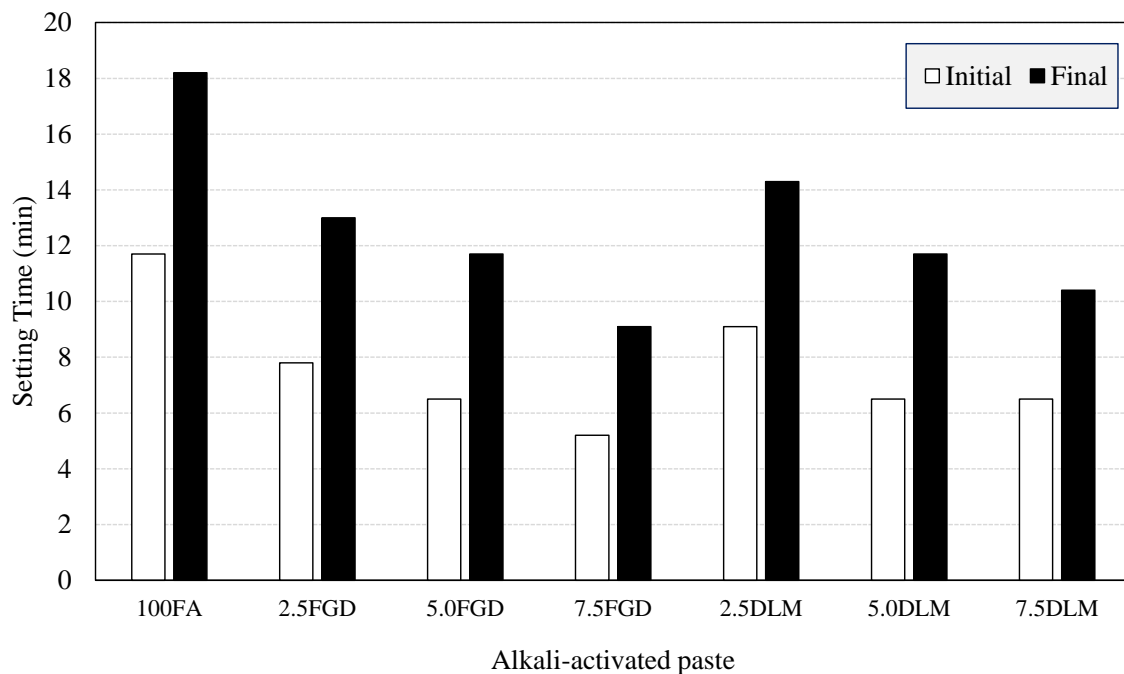


Figure 10 Setting time of alkali-activated FA paste under the best contents of FGD and DLM at L/B ratio of 0.40

3.3 Compressive strength

Figures 11-12 show the results of the compressive strength of the AAFA paste using FGD and DLM as the expansive additive. The compressive strength of the AAFA pastes tends to increase with the increase of FGD until a turning point. For example, 28-day strengths of 100FA, 2.5FGD, 5.0FGD, and 7.5FGD are 45.9, 50.2, 54.2, and 48.8 MPa, whereas 90-day strengths are 53.2, 57.9, 59.7, and 56.9 MPa, respectively. The strength increase of the AAFA containing FGD is due to the CaO oxide from FGD that could react with SiO_2 and Al_2O_3 from FA and hence become C-S-H and/or C-A-S-H. Besides, the Ca(OH)_2 from reaction products (see Eq. (1)) reacts with SiO_2 and/or Al_2O_3 from FA to form pozzolanic reaction like the blended cement paste. Similarly, the dissolution of Al^{3+} ions from FA increased by the reaction of SO_4^{2-} ions as reported by Boonserm et al. [50] and Ma et al. [51]. Differences in the trend of strength development of the AAFA paste with FGD could observe at the curing times of 180 days, implying that the strength development of the paste with FGD starts to decline compared to those of 120-day strength. For example, 120-day strengths of 100FA, 2.5FGD, 5.0FGD, and 7.5FGD are 58.0, 60.8, 63.1, and 60.8 MPa, whereas 180-day strengths are 60.2, 60.7, 60.3, and 55.6 MPa, respectively. As noticed in Eq. (1), FGD consists of sodium sulfate (Na_2SO_4); therefore, this reaction may further cause structural degradation of specimens. Also, ettringite formation ($\text{Ca}_6\text{Al}_2(\text{SO}_4)_3(\text{OH})_{12} \cdot 26\text{H}_2\text{O}$) had adverse effects on the specimens of AAFA paste, causing the expansion and/or cracking of specimens [38] and hence strength reduction.

Regarding the effect of using DLM in the AAFA paste, it is found that the increasing replacement levels of DLM enhance the strength development of the AAFA paste until the curing time of 180 days. For example, 120-day strengths of 2.5DLM, 5.0DLM, and 7.5DLM are 48.0, 53.3, and 55.9, whereas 180-day strengths are 64.5, 68.8, and 66.5 MPa, respectively. The noticeable increase in strength development is because the presence of CaO oxide from DLM assists the hardening with the addition of C-S-H and/or C-A-S-H, and coexisted with

N-A-S-H or geopolymer gel [20, 52]. Furthermore, Yip et al. [25] claimed that the Mg^{2+} ions played a similar role to Ca^{2+} ions, implying that MgO from DLM reacts with SiO_2 from FA to form magnesium silicate hydrate (M-S-H). The existence of M-S-H and C-S-H within the matrix contributed to the strength development of the AAFA paste [27, 28]. In addition, the chemical reaction of CO_3^{2-} ions in NaOH solution became calcium carbonate ($CaCO_3$), resulting in filling up the pore structure as reported by Ye et al. [53]. According to Figure 12, it is found that the strength development of AAFA paste incorporated with 7.5%DLM tends to drop compared to that of 5.0%DLM since the curing time of 120 days. Yip et al. [25] demonstrated that a small amount of carbonate material could enhance the strength development of geopolymer binder at the later stage. As reported by Yip et al. [25], mixing of 20% calcite and 20% dolomite in metakaolin-based geopolymers could enhance its strength development. However, in this study, the use of 5% DLM provides the high performance of the AAFA paste. The difference in precursors used for producing alkali-activated binders is an important reason. Therefore, the recommended expansive additive using in AAFA paste is DLM, which provides the long-term strength and low shrinkage of the AAFA paste, although 7.5%DLM is still higher than that of the 100FA paste.

From the test results, it can be summarized that the use of 5.0%FGD and 7.5%DLM as the expansive additive activated with the L/B ratio of 0.40 for producing the AAFA paste has been selected. This is because they show the highest 90-day compressive strength of the AAFA paste as desired, although the use of 5%DLM is advantageous in term of long-term strength. For this reason, they were selected to investigate the experiments of XRD, SEM/EDS, and FTIR for an observation of the development of geopolymerization products and to understand the effect of expansive additive.

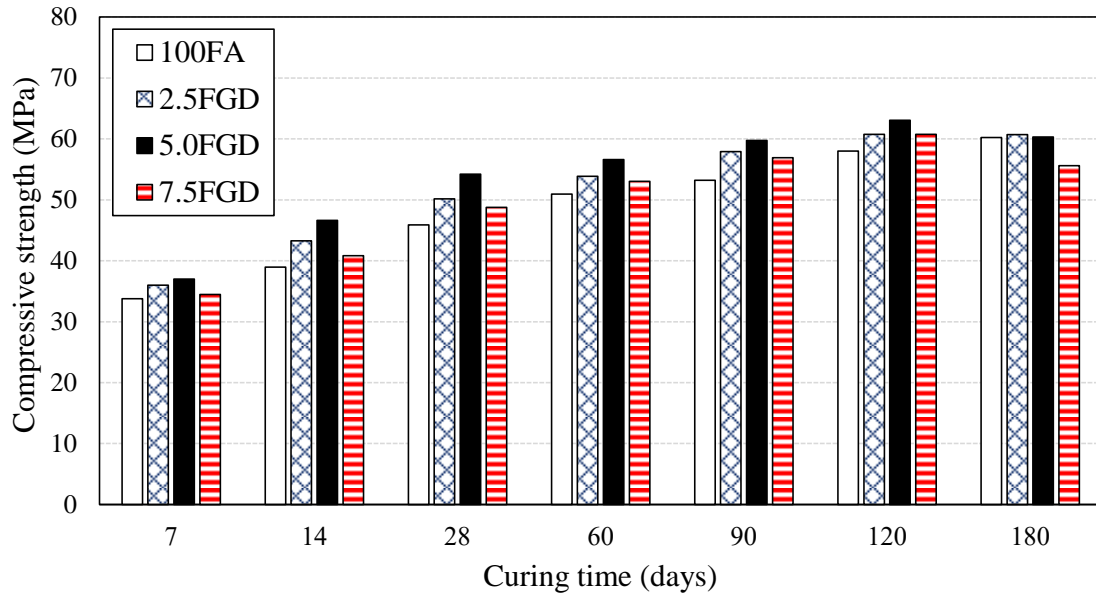


Figure 11 Compressive strength of alkali-activated FA paste under FGD contents at L/B ratio of 0.40

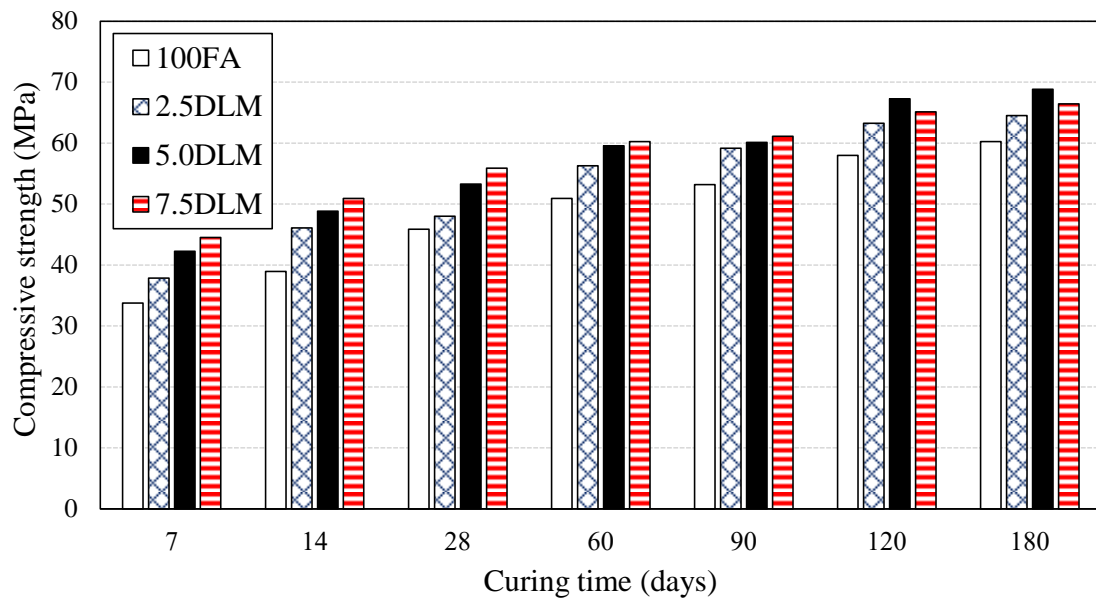


Figure 12 Compressive strength of alkali-activated FA paste under DLM contents at L/B ratio of 0.40

3.4 XRD analysis

The XRD patterns of the AAFA paste incorporated with FGD and DLM as the expansive additive at curing time of 90 days are shown in Figure 13. The AAFA paste without

expansive additive (Figure 13a) consists of amorphous phase at the hump around $25-40^{\circ}$ 2θ and crystalline phases of quartz (SiO_2), ferric oxide (Fe_2O_3), calcium carbonate (CaCO_3), and calcium silicate hydrate (C-S-H). The presence of C-S-H in the XRD pattern associated to the strength development of high-calcium based geopolymer which coexisted with N-A-S-H gel [16, 34, 46, 54]. When the paste containing FGD and DLM, the XRD patterns of 5.0%FGD (Figure 13b) and 7.5%DLM (Figure 13c) still consist of amorphous phase at the hump around $25-40^{\circ}$ 2θ and crystalline phases of quartz (SiO_2), ferric oxide (Fe_2O_3), calcium carbonate (CaCO_3), and calcium silicate hydrate (C-S-H). According to Figures 13b and 13c, the broad peak at 30° 2θ tends to be evident when compared to that of the 100FA paste. Moreover, the peaks of calcium silicate hydrate (C-S-H) and calcium carbonate (CaCO_3) are obviously evident. This is why its strength development increases.

A comparison between FGD and DLM is investigated and it is found that the peaks of calcium silicate hydrate (C-S-H) and calcium carbonate (CaCO_3) of the AAFA paste with DLM at 30° 2θ are higher than those of the AAFA paste with FGD. This is because the DLM mainly consists of calcium carbonate (CaCO_3) and dolomite ($\text{CaMg}(\text{CO}_3)_2$). Also, the reaction between dolomite ($\text{CaMg}(\text{CO}_3)_2$) and NaOH solution could produce the formation of calcium carbonate (CaCO_3) (see Eq. (2)) as reported by Garcia et al. [39] and Tong and Tang [55]. In addition, sodium carbonate (Na_2CO_3) obtained from Eq. (2) could react with calcium hydroxide ($\text{Ca}(\text{OH})_2$) from hydration products to form the formation of calcium carbonate (CaCO_3) [39]. The occurrence of high calcium carbonate (CaCO_3) content may be advantageous regarding the strength development and drying shrinkage of the AAFA paste with DLM at the later stage.

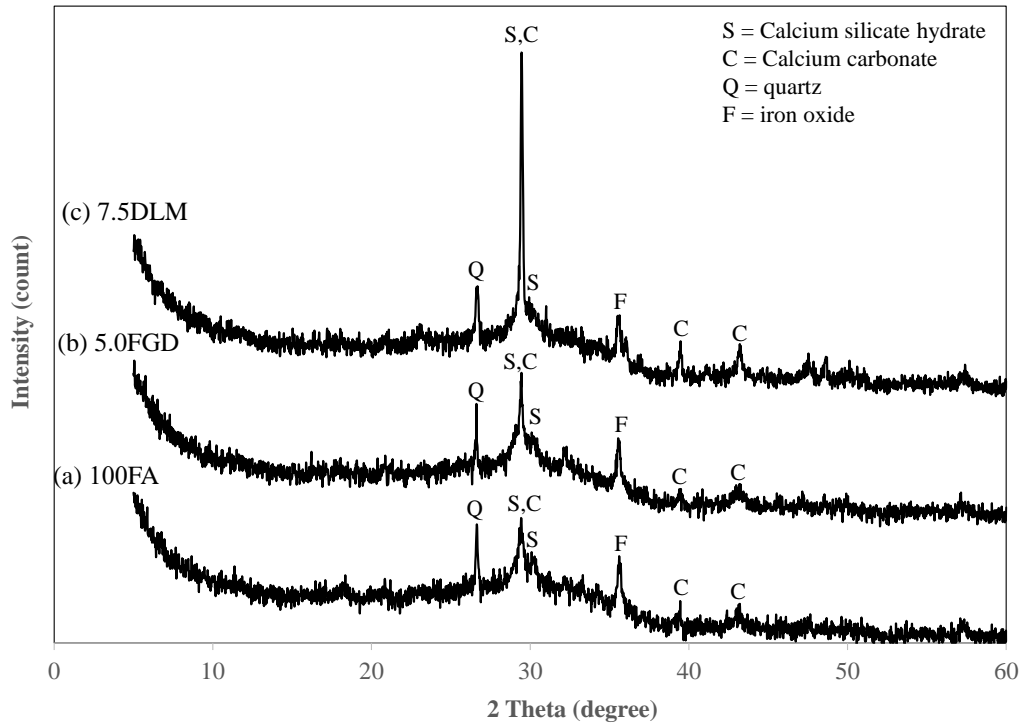


Figure 13 XRD patterns of alkali-activated FA paste containing FGD and DLM at 90 days

3.5 FTIR analysis

Figure 14 shows the results of FTIR spectroscopic of the AAFA paste with FGD and DLM as the expansive additive at curing time of 90 days. The strength development of alkali-activated binder can be explained by the degree of geopolymerization via the FTIR analysis. The considerable broad band located at $3700\text{--}2200\text{ cm}^{-1}$ and $1700\text{--}1600\text{ cm}^{-1}$ were assigned to O-H stretching and O-H bending of water molecule, respectively, as reported by Panias et al. [56] and Boonserm et al. [50]. The spectra located at $1400\text{--}1450\text{ cm}^{-1}$ represented the carbonate (CO_3^{2-}) band [57], whereas a combination of C-S-H and N-A-S-H gel was located at 900 and 1200 cm^{-1} [57-59]. In addition, the unreacted SiO_2 in the matrix was located at approximately 460 cm^{-1} [60].

According to Figure 14, the wave number at approximately 3350 cm^{-1} and 1640 cm^{-1} had marginal change, implying that the precursors reacted with an alkali solution as expected.

Similarly, the 100FA, 5.0FGD, and 7.5DLM pastes had marginal change regarding the degree of geopolymerization in the matrix as observed at approximately 945 cm^{-1} . This is because these mixes were selected based on low shrinkage and highest strength as mentioned previously. However, at the wave number of approximately 460 cm^{-1} , the unreacted SiO_2 of the 7.5DLM paste exhibits less than those of the 100FA and 5.0FGD pastes. This behavior corresponds to the results of XRD patterns (see Figure 13). On the other hand, the 7.5DLM paste is easily identified which is related to the peak of calcium carbonate (CaCO_3) in XRD results (see Figure 13c). Yip et al. [25] claimed that the use of a small amount of dolomite ($\text{CaMg}(\text{CO}_3)_2$) for producing the alkali-activated binders could improve the long-term strength and hence the carbonate (CO_3^{2-}) might easily be detected in the matrix. According to Figure 14, the SO_4^{2-} bonding has been detected at approximately 600 cm^{-1} for the paste with FGD (Figure 14b). This result agrees with the work of Boonserm et al. [50], who reported the improved geopolymerization of bottom ash and FA by using FGD as an additive. The presence of SO_4^{2-} bonding has adverse effects on the strength development of the AAFA paste with FGD at the curing time of 120 days (see Figure 11).

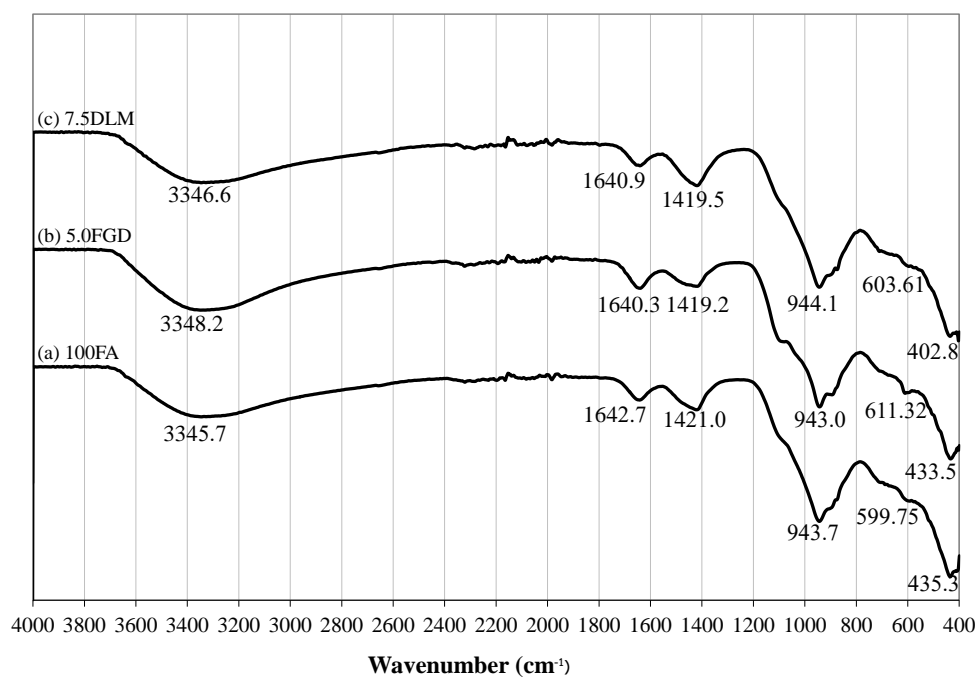


Figure 14 FTIR spectra of alkali-activated FA paste containing FGD and DLM at 90 days

3.6 SEM and EDS analysis

The results of SEM and EDS images of the AAFA paste with FGD and DLM as the expansive additive at curing time of 90 days are illustrated in Figures 15-16. All samples exhibit gel formation adjacent FA particles and some non-reacted and/or partially reacted FA particles embedded in a continuous matrix. The existence of calcium oxide in the system results in the development of C-S-H coexisted with N-A-S-H gel which is related to peaks of C-S-H in the XRD patterns. According to Figure 15, the AAFA pastes incorporated with FGD (Figure 15c) and DLM (Figure 15e) seem to be denser than that of the AAFA paste without the expansive additive (Figure 15a). As shown in Figure 15b, the regular-shape product of C-S-H is obviously seen on the FA particles of the 100FA paste. For the 5.0FGD paste (Figure 15d), a large amount of needle-shape products, known as the formation of “ettringite”, is obviously noticed; however, the formation of ettringite is difficult to be detected by the XRD pattern [37]. Chinadaprasirt et al. [37] claimed that the formations of calcium aluminate monosulfate ($3\text{CaO} \cdot \text{Al}_2\text{O}_3 \cdot \text{CaSO}_4 \cdot 12\text{H}_2\text{O}$) and calcium sulfate (CaSO_4) would be obtained from the formation of ettringite ($\text{Ca}_6\text{Al}_2(\text{SO}_4)_3(\text{OH})_{12} \cdot 26\text{H}_2\text{O}$) in the matrix. Thus, it is a negative factor for the strength development of the AAFA paste at the later stage. For the paste with 7.5DLM (Figure 15f), the reaction products obtained from the dedolomitization process in alkali solution were formed, and hence they coexisted with those of C-S-H, C-A-S-H, and N-A-S-H gel [61]. As seen from Figure 15f, the appearance of calcium carbonate (CaCO_3) and magnesium hydroxide ($\text{Mg}(\text{OH})_2$) is obviously seen on the FA particles. In addition, the formation of C-S-H gel is also observed. This could explain why they had enhanced strength development.

In order to understand the reaction products within the matrix, the EDS analysis was performed on the 100FA, 5.0FGD, and 7.5DLM pastes. It is indicated that the main elements in the products viz., Na, Mg, Al, Si, Ca, and Fe are detected. The presence of Na, Si, and Al confirm that N-A-S-H gel exists in the matrix, which is produced from the geopolymerization process. Likewise, the existence of Ca, Si, and Al represented the reaction products of C-S-H and C-A-S-H gel. For the 100FA paste, it has the Si/Al ratios of 2.02, 1.82, 3.47, and 0.29 for spectrum 1, 2, 3, and 4, respectively. Besides, the Ca/Si ratio of the 100FA paste exhibits a low value, which is related to its chemical composition (see Table 1). Song et al. [62] demonstrated that the low Ca/Si ratio represented the beginning reaction between Ca^{2+} and Si^{4+} ions in alkali solutions to form C-S-H gel. When the paste contains FGD and DLM, the variations in Ca/Si ratio are obtained. The Ca/Si ratios ranged between 0.84-0.91 and 0.71-1.25 for FA-FGD and FA-DLM pastes, respectively. This is in agreement with the studies of Gue et al. [63] and Li et al. [15] that the Ca/Si ratio at the beginning hydration was approximately 1.0. As revealed in Figures 16b and 16c, the Si/Al ratios vary between 2.16-2.33 and 2.88-2.91 for FA-FGD and FA-DLM pastes, respectively. The Ca/Si ratio of the AAFA paste with DLM agreed with the works of Punurai et al. [20] and Gue et al. [41]. They claimed that the Si/Al ratio in the ranges of 2.6 - 3.0 provided a high strength development of high-calcium based geopolymer. According to spectrum 1 in Figure 16b, a high amount of Al was found; therefore, ettringite formation had taken place, in which the needle-shape products occurred. For spectrum 1 in Figure 16c, the elements of Mg and Ca contain a high amount that corresponds to a high ratio of Ca/Si. In this case, the reaction products of dedolomitization participated in the system, and hence it might be formations of calcium carbonate (CaCO_3) and magnesium hydroxide (Mg(OH)_2). Considering the Na/Al ratio, they are 0.85, 0.65, and 1.07 for spectrums 3 (Figure 16a), 4 (Figure 16b), and 3 (Figure 16c), respectively. The paste with DLM (Na/Al ratio of 1.07) is consistent with the work of Zhang

et al. [64], who reported that the Na/Al ratio should be 1.0. The results of EDS corresponds well to the result of their strength.

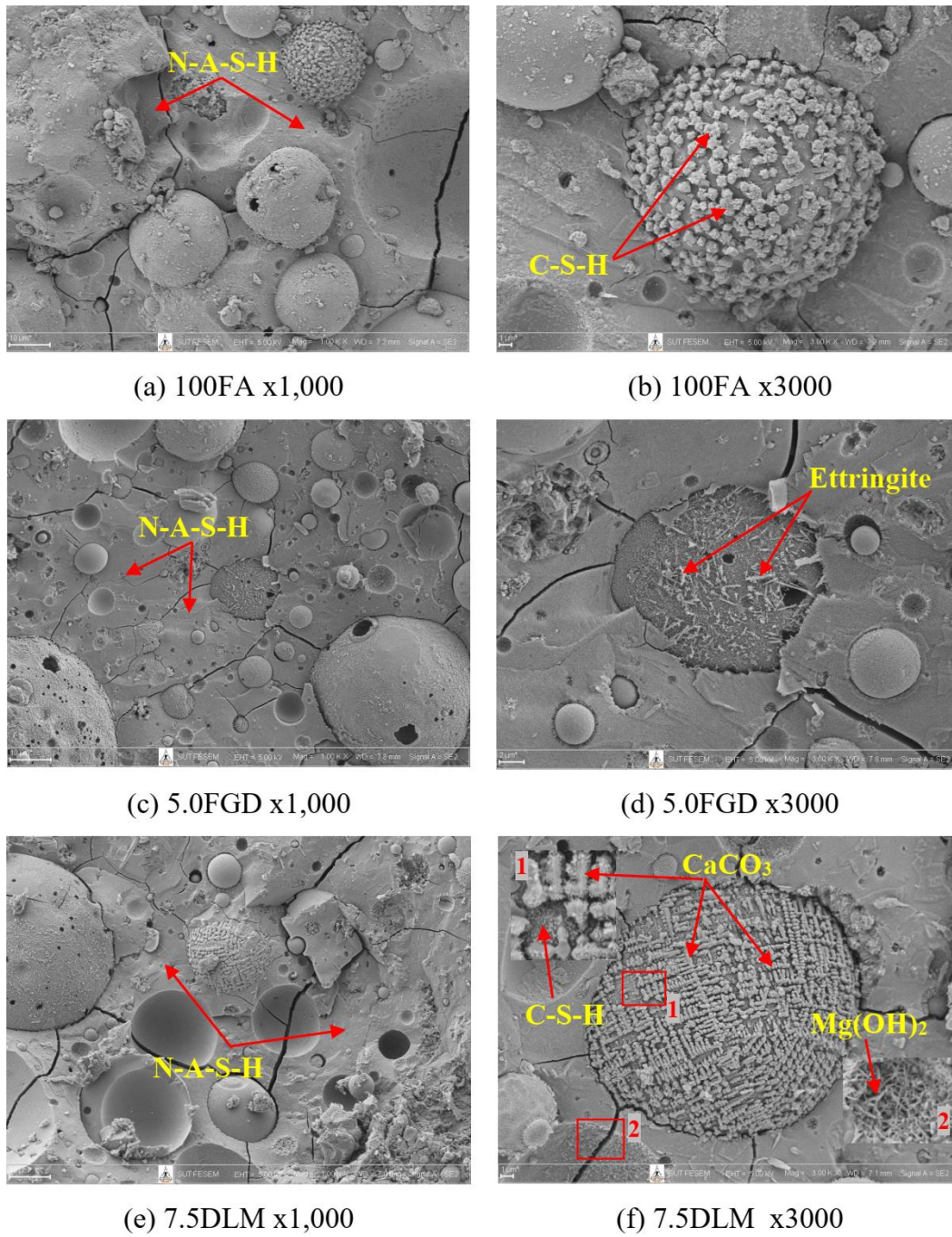


Figure 15 SEM images of alkali-activated FA paste containing FGD and DLM at 90 days

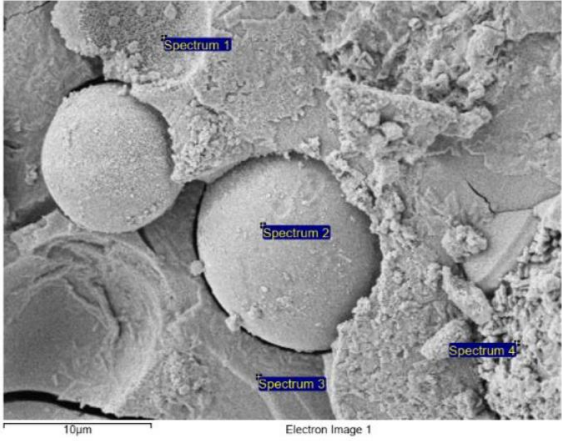
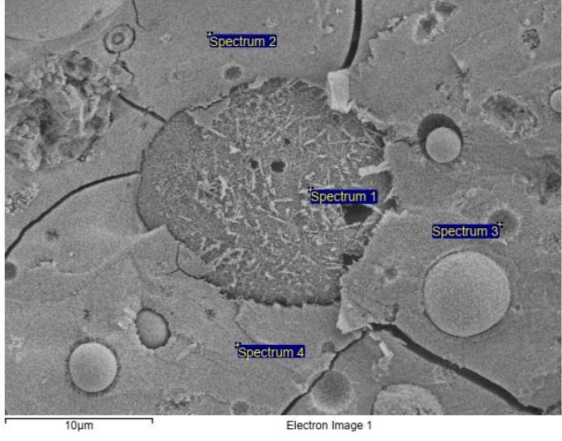
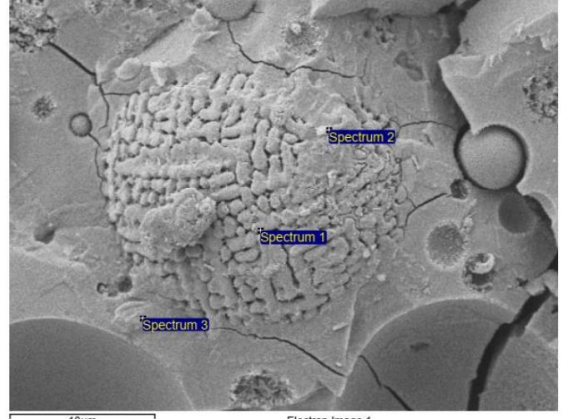
	Spectrum 1	Spectrum 2	Spectrum 4
(a) 100FA	Element	Atomic (%)	Atomic (%)
	O	62.19	61.29
	Na	2.94	1.13
	Mg	3.99	-
	Al	4.50	6.66
	Si	9.11	12.13
	Ca	3.31	3.59
	Fe	5.13	7.51
	Si/Al	2.02	1.82
	Ca/Si	0.36	0.30
	Na/Al	0.65	0.17
		Spectrum 1	Spectrum 3
(b) 5.0FGD	Element	Atomic (%)	Atomic (%)
	O	48.50	39.24
	Na	0.44	6.73
	Mg	1.74	4.14
	Al	23.94	9.72
	Si	9.43	22.69
	Ca	4.32	20.58
	Fe	4.99	1.90
	Si/Al	0.39	2.33
	Ca/Si	0.32	0.91
	Na/Al	0.02	0.69
		Spectrum 1	Spectrum 2
(c) 7.5DLM	Element	Atomic (%)	Atomic (%)
	O	62.81	53.51
	Na	0.94	1.77
	Mg	9.95	-
	Al	2.27	2.43
	Si	1.15	7.08
	Ca	16.08	8.87
	Fe	6.81	3.97
	Si/Al	0.50	2.91
	Ca/Si	13.98	1.25
	Na/Al	0.41	0.73
		Spectrum 2	Spectrum 3

Figure 16 EDS images of alkali-activated FA paste containing FGD and DLM at 90 days

4. Conclusion

This paper presents an experimental study on the drying shrinkage, strength, and microstructure of alkali-activated high-calcium FA using FGD-gypsum and dolomite as the expansive additive. Based on the obtained experimental results, the following conclusions could be drawn:

1) The drying shrinkage of alkali-activated FA paste is significantly improved with the incorporation of FGD and DLM as the expansive additive. While the increase in the L/B ratios leads to increased drying shrinkage of alkali-activated FA paste for all replacement levels of FGD and DLM. The reduction of drying shrinkage of alkali-activated FA paste incorporated with FGD and DLM was because of the reaction products of gypsum and dolomite in alkali solution.

2) The incorporation of FGD and DLM in alkali-activated FA paste resulted in a reduction in its setting time, whereas the strength development tended to increase. However, a strength reduction was found at the later stage noticed in the paste with 5.0%FGD at the curing time of 120 days. The increase in strength was due to the addition of CaO oxide from FGD and DLM reacted with SiO_2 and Al_2O_3 from FA that accelerate the geopolymerization process; therefore, the speedy setting time and strength development of alkali-activated FA paste were obtained.

3) The main reaction products viz., amorphous phase, C-S-H, and calcite of alkali-activated FA paste with DLM in the XRD patterns exhibited higher value than those of the 100FA and 5.0FGD pastes. A small amount of dolomite used in the matrix could enhance the long-term strength and the existence of carbonate (CO_3^{2-}) may have positive effect on its strength development. For the paste with FGD, the SO_4^{2-} bonding was found in the FTIR analysis and this bonding caused a loss in strength development of paste seen at the curing time of 120 days.

4) The SEM images of alkali-activated FA paste with DLM and FDG showed an improved microstructure in comparison with the paste without the expansive additive. Different reaction products surrounding FA particles were noticed. This includes the formations of ettringite and C-S-H were observed for FA-FGD paste; whereas the formations of CaCO_3 , Mg(OH)_2 , and C-S-H were seen for FA-DLM paste, which coexisted with N-A-S-H gel. The results of EDS analysis correspond well to the results of SEM image and strength development.

5) The alkali-activated FA paste with DLM as the expansive additive is recommended for the optimum mix in terms of improved drying shrinkage and long-term strength compared with that of the paste with FGD.

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